

# Growth of Hydroxyapatite Coatings on Biodegradable Polymer Microspheres

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**ABSTRACT** Mineral-coated microspheres were prepared via a bioinspired, heterogeneous nucleation process at physiological temperature. Poly(D,L-lactide-co-glycolide) (PLG) microspheres were fabricated via a water-in-oil-in-water emulsion method and were mineral-coated via incubation in a modified simulated body fluid (mSBF). X-ray diffraction, Fourier transform infrared spectroscopy, and scanning electron microscopy with associated energy-dispersive X-ray spectroscopy confirmed the presence of a continuous mineral coating on the microspheres. The mineral grown on the PLG microsphere surface has characteristics analogous to those of bone mineral (termed “bonelike” mineral), with a carbonate-containing hydroxyapatite phase and a porous structure of platelike crystals at the nanometer scale. The assembly of mineral-coated microspheres into aggregates was observed when microsphere concentrations above 0.50 mg/mL were incubated in mSBF for 7 days, and the size of the aggregates was dependent on the microsphere concentration in solution. In vitro mineral dissolution studies performed in Tris-buffered saline confirmed that the mineral formed was resorbable. A surfactant additive (Tween 20) was incorporated into mSBF to gain insight into the mineral growth process, and Tween 20 not only prevented aggregation but also significantly inhibited mineral formation and influenced the characteristics of the mineral formed on the surface of PLG microspheres. Taken together, these findings indicate that mineral-coated PLG microspheres or mineral-coated microsphere aggregates can be synthesized in a controllable manner using a bioinspired process. These materials may be useful in a range of applications, including controlled drug delivery and biomolecule purification.

**KEYWORDS:** mineralization • PLG microsphere • bioinspired • drug delivery • HAP chromatography

## I. INTRODUCTION

Calcium phosphate-based mineral coatings are desirable interfaces for several biomedical applications because they are similar in composition to bone tissue and have been shown to be conducive to the ongrowth and/or ingrowth of newly formed bone. For example, hydroxyapatite, the major inorganic component of bone mineral, has been shown to be osteoconductive (1) and may also be capable of inducing new bone formation in vivo (2). Another important property of hydroxyapatite is its ability to bind with high affinity to biological molecules. For example, hydroxyapatite has been commonly used as a resin for chromatographic purification of proteins and plasmid DNA (3, 4) because the mineral surface contains both positive ( $\text{Ca}^{2+}$ ) and negative ( $\text{PO}_4^{3-}$ ) ions capable of binding electrostatically with basic and acidic biological molecules, respectively. More recently, the high affinity of hydroxyapatite for proteins has also been used as a mechanism to bind and release biologically active molecules, including albumin, BMP-2, TGF- $\beta$ 1, IGF-1, HGF, and bFGF (5–11). Therefore, a large body of work has focused on the creation of hydroxyapatite coatings on polymeric biomaterials to exploit the unique advantages of hydroxyapatite coatings.

A particular subset of approaches used to grow hydroxyapatite coatings on biomaterials mimics some aspects of natural biomineralization processes and has therefore been termed “biomimetic” or “bioinspired” (12–16). Kokubo et al. first reported bioinspired growth of apatite coatings on bioactive CaO–SiO<sub>2</sub> glass in a simulated body fluid (SBF), which had ion concentrations nearly equal to those of human blood plasma and was held at physiological temperature and pH (17). A series of subsequent studies reported mineral growth using novel formulations of SBF (18), variations in the mineral growth process (19), or variations in the base materials (20). The basis for mineral nucleation in these studies involved interactions of mineral ions in solution with polar functional groups on the materials surface, such as Si–OH (21), Ti–OH (22), Zr–OH (23). A series of recent studies have extended the bioinspired mineralization process to include the formation of bonelike hydroxyapatite coatings on biodegradable polymer films (24) or porous scaffolds (25–27). The mechanism for mineral nucleation and growth on these materials is based on the interaction of carboxylate ions and hydroxyl groups on the hydrolyzed surface with calcium- and phosphate-rich nuclei in solution (24). This coating process is particularly suitable for biodegradable polymers because it can be carried out at physiological temperature and pH (28), and the mild processing conditions have also allowed investigators to coprecipitate minerals with biologically active proteins during the coating process (29–33). Therefore, bioinspired growth of mineral coatings may be particularly advantageous in drug delivery applications, enabling protein to be incorporated and re-

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leased while maintaining the biological activity of a released protein (8, 29, 34–36).

In view of the potentially advantageous properties of bioinspired hydroxyapatite coatings, we have developed a bioinspired process to grow hydroxyapatite mineral coatings on the surface of biodegradable polymer microspheres, which are a popular platform in contemporary drug delivery applications. We hypothesized that these microspheres, which have been previously shown to possess a negatively charged surface, would provide a template for bioinspired nucleation and growth of hydroxyapatite. The resulting microspheres are designed to exploit the controllable properties of polymer microspheres (e.g., size, size range, aggregation, degradability, and drug incorporation), while also taking advantage of the biological properties of the mineral layer (e.g., biomolecule binding/incorporation). Our processing method had two steps: (1) poly(D,L-lactide-co-glycolide) (PLG) microspheres were synthesized using a double emulsion method (37); (2) microspheres were incubated in a modified SBF (mSBF), allowing for mineral nucleation and growth in near-physiological conditions. A recent communication from our group demonstrated that mineral coatings can be grown on PLG microspheres via this process and that these coatings can serve as platforms for protein binding and release. The purpose of this more detailed study is to provide a thorough characterization of the mineral growth process and the properties of the resulting mineral layer. In addition, the characteristics of mineral-coated microspheres are modulated in the current study by varying the incubation time, the microsphere concentration, and the presence of surfactant during the coating process. The results presented here illustrate that an inorganic mineral layer can be grown in a controllable manner on the surface of biodegradable microspheres, and these materials may find utility in applications such as drug delivery and protein chromatography.

## II. EXPERIMENTAL SECTION

**a. Microsphere Fabrication and Characterization.** 85:15 PLG (average  $M_w = 50\,000$ – $70\,000$ ) and poly(vinyl alcohol) (PVA; MW 9–10 kDa) were obtained from Sigma-Aldrich (St. Louis, MO). All chemicals and solvents were of reagent grade and were obtained from Fisher Chemicals (Fair Lawn, NJ).

PLG microspheres were fabricated by water-in-oil-in-water (W/O/W) double-emulsion technique as reported elsewhere (13, 38). Briefly, the organic phase consisted of 5% (w/v) PLG in 1 mL of ethyl acetate. The aqueous phase consisted of 0.1 mL of phosphate-buffered saline (PBS). The aqueous and organic phases were mixed and sonicated using a Sonifier 250 (VWR International, Inc., West Chester, PA) for 15 s. The resulting first emulsion was added immediately into 1 mL of aqueous 1% (w/v) PVA in 7% (v/v) ethyl acetate that was being mechanically vortexed for 15 s to form a second emulsion. The resulting solution was then added to a beaker containing 200 mL of 0.3% PVA in 7% ethyl acetate and further rigorously stirred for 4 h to allow for organic solvent evaporation. The resulting microspheres were collected by filtration through a 0.22  $\mu\text{m}$  filter, washed three times with deionized water, and resuspended in deionized water. The microspheres were lyophilized for a minimum of 48 h and were stored at  $-20\text{ }^\circ\text{C}$  in the presence of a desiccant.

To confirm that PLG microspheres were negatively charged in physiological buffers, we first characterized the  $\zeta$  potential of PLG microspheres in PBS and mSBF solutions. The surface charge of the microsphere particles was measured with a Zetasizer 3000HS (Malvern Instruments, Worcestershire, U.K.). The electrophoretic mobility of uncoated microspheres in three 6 mL aliquots was measured to determine the surface potential, with each injection having five measurements. Samples were syringe-loaded and measured at  $25\text{ }^\circ\text{C}$  in 1xPBS or mSBF, at a pH of 6.8 to mimic mineral coating conditions.

Quantification of aggregated microspheres in various buffers was performed by incubating a 0.5% (w/v) PLG microsphere in a selected buffer (1xPBS, mSBF, and mSBF + 0.1% (v/v) Tween 20) for 1, 3, and 7 days. The suspension was held at  $37\text{ }^\circ\text{C}$  and rotated continuously for the duration of the study period, identical with the conditions used for mineral growth. Prior to changing the buffer on the subsequent day, aliquots of each condition were taken, diluted 1:8, and imaged under an Olympus Ix51 light microscope at  $20\times$  magnification. Four photographs were taken per sample per time point with a Hamamatsu 1394 ORCA-285 camera. The resultant images were viewed and counted using *Image J* software.

**b. Mineral Coating of Microspheres.** PLG microspheres were coated with a mineral layer via incubation in mSBF for 7 days. The mSBF solution was replaced daily to ensure adequate ion concentrations for mineral growth. mSBF possesses inorganic ion concentrations similar to those of human blood plasma, with 2 times the concentration of calcium and phosphate ions. mSBF was prepared by dissolving 141 mM NaCl, 4.0 mM KCl, 0.5 mM  $\text{MgSO}_4$ , 1.0 mM  $\text{MgCl}_2$ , 4.2 mM  $\text{NaHCO}_3$ , 5.0 mM  $\text{CaCl}_2$ , and 2.0 mM  $\text{KH}_2\text{PO}_4$  in distilled water, buffered to pH 6.8, and was held at  $37\text{ }^\circ\text{C}$  for the duration of the incubation period. In some experiments, 0.1% Tween 20 (Sigma-Aldrich, St. Louis, MO) was added to mSBF to influence the aggregation of microspheres.

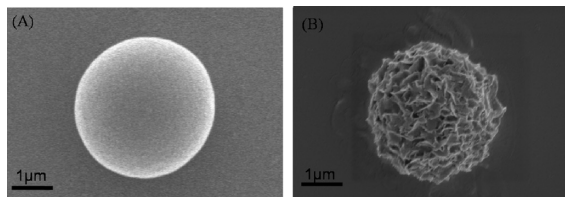
**c. Materials Characterization.** The composition and phase of the minerals grown on polymer microspheres were analyzed using a HI-STAR two-dimensional X-ray diffractometer (Siemens Corp., New York, NY) operating at 40 kV and 20 mA. X-ray diffraction (XRD) spectra were taken for  $2\theta = 20$ – $40^\circ$ , and data collection was controlled using General Area Detector Diffraction System version 4.0 (Bruker AXS Inc., Madison, WI).

Fourier transform infrared (FTIR) spectroscopy data were obtained using a EQUINOX 55 spectrometer (Bruker AXS Inc., Madison, WI). Samples were examined in transmission mode in the  $400$ – $4000\text{ cm}^{-1}$  range, and data were analyzed by *OPUS* software.

The morphology and composition of the coated mineral on the microsphere surface was analyzed using scanning electron microscopy (SEM) with energy-dispersive X-ray spectroscopy (EDS). Microspheres before and after mSBF incubation were mounted on aluminum stubs with double-sided carbon tape, sputtered with gold for 30 s at 45 mA, and characterized using a LEO DSM 1530 field-emission scanning electron microscope, operating at 2 kV for SEM and 10 kV for EDS.

Dissolution of mineral coatings was characterized by measuring the release of  $\text{PO}_4^{3-}$  and  $\text{Ca}^{2+}$  over 25 days in multiple solutions, including Tris-buffered saline (TBS; 150 mM NaCl and 20 mM Tris, pH = 7.4) or Dulbecco's modified eagle medium (DMEM) without L-glutamine and phenol red (Mediatech, Inc., Manassas, VA). A total of 5 mg of mineral-coated microspheres was immersed in 1 mL of a buffer solution and incubated with continuous rotation for 24 h at  $37\text{ }^\circ\text{C}$ . The buffer was collected and refreshed daily. The study was performed in triplicate and held at  $37\text{ }^\circ\text{C}$  for the duration of the dissolution study period.

The amount of phosphate released from mineral-coated microspheres was analyzed colorimetrically using an assay previously reported (39). Briefly, a working acetone–acid–molybdate (AAM) solution was prepared by mixing 2 parts of



**FIGURE 1.** PLG microsphere (A) and mineral-coated microsphere after incubation in mSBF for 7 days (B).

acetone with 1 part of 5 N sulfuric acid, and 1 part of a 10 mM ammonium molybdate solution. The assay was performed in a 96-well plate by adding 100  $\mu\text{L}$  of a freshly made working solution to 100  $\mu\text{L}$  of the sample. The amount of phosphate complex was quantitatively detected by measuring the absorbance at 405 nm on a Synergy HT Multidetector Microplate Reader (BioTek Instruments Ltd., Bedfordshire, U.K.) and comparing it to a set of standards with known phosphate concentrations.

$\text{Ca}^{2+}$  release was directly measured using a QuantiChrom Calcium Assay Kit (DICA-500; BioAssay Systems, Hayward, CA). A phenolsulfonephthalein dye forms a very stable blue color complex with free calcium. The intensity of the complex, measured via absorbance at 612 nm, was used to measure released  $\text{Ca}^{2+}$  by comparing it to a set of standards with known calcium concentrations. Dissolution experiments were performed in triplicate, and the average of calcium and phosphate releases was reported.

### III. RESULTS AND DISCUSSION

**Fabrication and Characterization of PLG Microspheres.** A double-emulsion process detailed in previous studies (37, 38) was used here to produce spherical particles of 85:15 PLG (Figure 1A). The  $\zeta$  potential of these particles in PBS ( $-81.09 \pm 7.71$  mV) and mSBF ( $-78.62 \pm 15.91$  mV) indicated that the particles were negatively charged (Figure 2A). These  $\zeta$  potential values are consistent with previous studies of PLG microspheres, which have also shown that PLG microspheres have negatively charged surface carboxylate groups (40) and that they have  $\zeta$  potential values ranging from  $-22$  to  $-80$ , depending on the microsphere preparation technique and the testing buffer (41–44). The presence of carboxylate groups on the surface of these microspheres is important because previous studies have indicated that these groups are capable of promoting heterogeneous mineral nucleation and growth. However, in previous studies, PLG materials were hydrolyzed to produce surfaces containing carboxylate ions, while in this case, the PLG microsphere surfaces included negatively charged groups when synthesized via double-emulsion processing without additional hydrolysis. Interestingly, the  $\zeta$  potential of PLG microspheres was significantly lower than that of hydrolyzed PLG films used previously as templates for bioinspired mineral nucleation and growth (Figure 2B) (24), suggesting that the microspheres may serve as advantageous templates for mineral growth.

**Mineral Nucleation and Growth.** Incubation of PLG microspheres in mSBF led to nucleation and growth of a hydroxyapatite mineral coating on the microsphere surface (Figure 1B). SEM observation showed that the nanocrystallites grown on the microsphere surface exhibit a platelike morphology (Figure 3), similar to the morphology observed in native bone tissue and in previous bioinspired mineral

growth studies (31, 45). Micrographs of the microspheres incubated in mSBF for 7 days show continuous mineral coatings on individual microspheres incubated at 0.25% and 0.50% (w/v) (Figure 3A,B). Microsphere aggregation was observed as the microsphere concentration increased, and mineral coatings were also observed on the surface of microsphere aggregates (Figure 3C,D).  $\zeta$  potential results showed charged microspheres in all buffers tested (Figure 2A), so it is possible that the presence of salt leads to shielding of the microsphere surface charge, thereby limiting electrostatic repulsion and facilitating aggregation. The size of the aggregates increases with the microsphere concentration because the size of the microsphere aggregates was proportional to the initial microsphere concentration in solution (Figure 3E). To determine whether the aggregation of the microspheres was due to some intrinsic property of mSBF, we performed aggregation experiments in which microspheres were incubated in three different solutions: (1) a 1xPBS solution (calcium-deficient), (2) mSBF, and (3) mSBF in the presence of Tween20. Each incubation was performed at 37  $^{\circ}\text{C}$  with daily solution changes for 7 days. Results showed that 35% and 42% microspheres aggregated after the first day of incubation in PBS and mSBF, respectively. The percentage of aggregated microspheres increased to 87% in PBS and 90% in mSBF after 7 days of incubation. In contrast, the number of microspheres aggregated in mSBF supplemented with Tween20 was significantly lower at each time point (Supplementary Figure 1 in the Supporting Information). These data indicate that mSBF does not significantly increase microsphere aggregation when compared to PBS and the presence of Tween20 significantly decreases microsphere aggregation.

**Characteristics of Mineral Coatings.** The phase and composition of mineral coatings on PLG microspheres after a 7 day incubation in mSBF were characterized by XRD and FTIR. XRD spectra of mineral-coated microspheres show three characteristic hydroxyapatite peaks at  $2\theta = 25.87^{\circ}$ ,  $28.68^{\circ}$ , and  $32.05^{\circ}$  similar to the peaks present in the XRD spectrum of a reagent-grade hydroxyapatite powder (Sigma-Aldrich, St. Louis, MO) at  $2\theta = 26^{\circ}$ ,  $28.5^{\circ}$ , and  $32^{\circ}$  (Figure 4A). The peak areas in the XRD spectrum of mineral-coated microspheres are broader than that of the hydroxyapatite powder, and this may be due to the small crystal size of the mineral deposited on the PLG microsphere surfaces. FTIR peaks observed in the 1600–400  $\text{cm}^{-1}$  region can be attributed to carbonate-substituted hydroxyapatite, including phosphate peaks at 570, 950, 1046, and 1098  $\text{cm}^{-1}$  and carbonate peaks at 870, 1410, and 1456  $\text{cm}^{-1}$ . The EDS spectrum also confirms the presence of calcium and phosphorus on the mineral-coated microspheres (Figure 4C). The Ca/P ratio of the mineral coating was 1.41 after 7 days of incubation in mSBF, which is consistent with that of previous studies of biological apatites (46) and bioinspired mineral coatings (45).

Time-lapse SEM analyses of PLG microspheres during mSBF incubation provide some insight into the mechanism of nucleation and growth of carbonate apatite mineral on

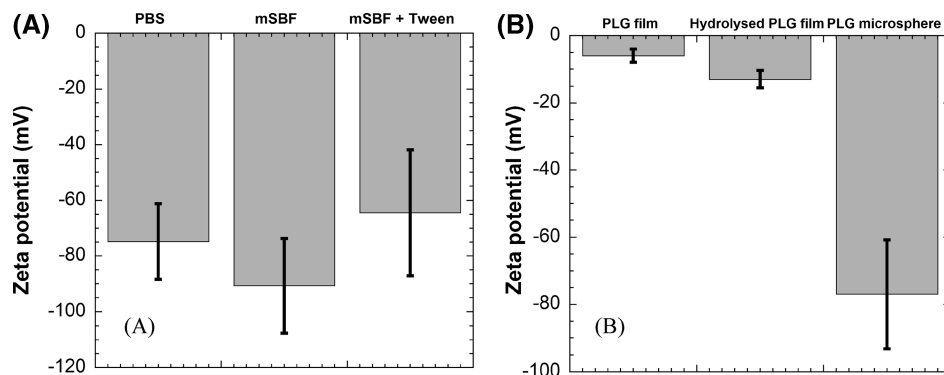


FIGURE 2.  $\zeta$  potential of (A) PLG microspheres in buffers PBS, mSBF, and mSBF + 0.1% (v/v) Tween20 and (B) nonhydrolyzed and hydrolyzed PLG films in comparison with PLG microspheres.

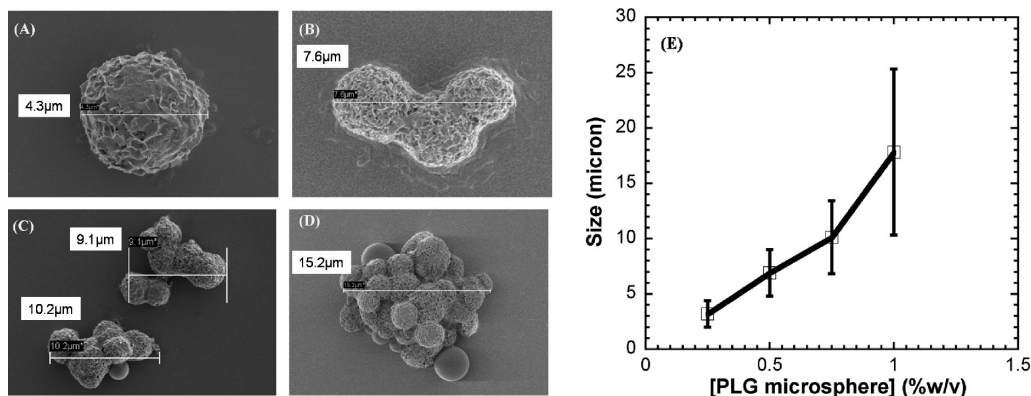


FIGURE 3. SEM images of mineral-coated microspheres after a 7 day incubation in a mSBF solution: (A) 0.25% (w/v); (B) 0.50% (w/v); (C) 0.75% (w/v); (D) 1.00% (w/v). (E) Relationship between the microsphere concentration in solution during mineral growth and the size of mineral-coated microsphere aggregates.

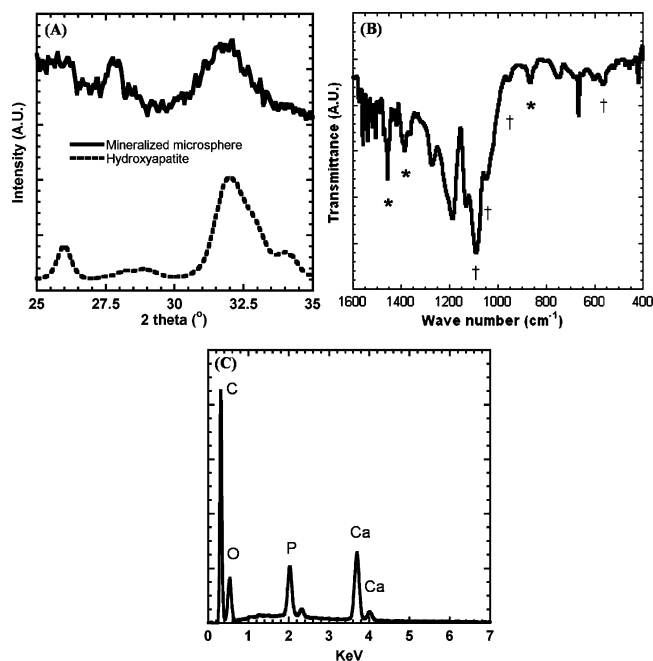


FIGURE 4. (A) XRD analysis of mineral-coated microspheres and a hydroxyapatite powder (included for comparison). (B) FTIR analysis of mineral-coated microspheres. Peaks associated with carbonate are denoted by asterisks, and peaks associated with phosphate are denoted by daggers. (C) EDS spectrum of mineral-coated microspheres after a 7 day incubation in mSBF.

aggregating microspheres (Figure 5). The nucleation process begins during the first 3 days of incubation in mSBF (Figure

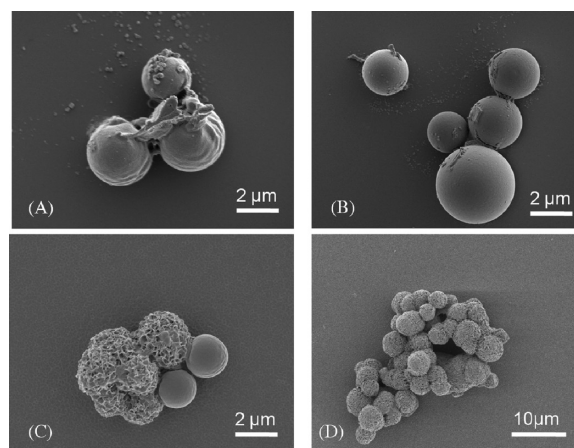


FIGURE 5. Process of mineral nucleation and growth on PLG microspheres. SEM images of microspheres after (A) the first day of immersion in mSBF, (B) day 3 of incubation, (C) day 5 of incubation, and (D) day 7 of incubation.

5A,B). During this stage, the aggregation of microsphere begins to occur. As the microspheres begin to aggregate, small crystals ( $\sim 2\text{--}10$  nm) begin to form at the interface between microspheres (Figure 5A,B), perhaps, in part, because of the local supersaturation of functional groups at adjacent microsphere interfaces, and associated mineral ions at the interfaces. Our data also suggest that aggregation is not a prerequisite for mineral nucleation and growth because at low concentrations of microspheres in solution [0.25–0.50% (w/v)] we observed individual microspheres

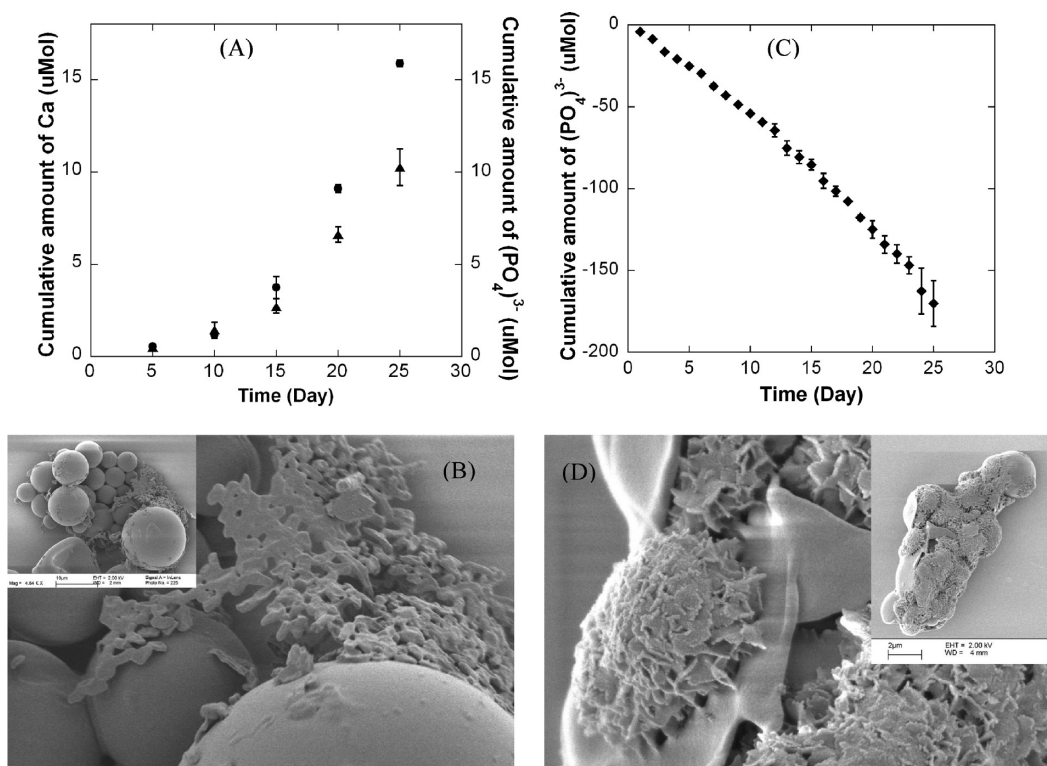


FIGURE 6. (A) Cumulative dissolution of  $\text{Ca}^{2+}$  and  $\text{PO}_4^{3-}$  during a 25 day incubation in TBS and (B) SEM images of mineral-coated microspheres after the 25 day TBS incubation. (C) Cumulative dissolution of  $\text{PO}_4^{3-}$  during a 25 day incubation in DMEM and (D) SEM images of mineral-coated microspheres after the 25 day DMEM incubation.

coated with mineral (Figure 3A,B). However, aggregation could have a positive impact on mineralization, and the processes could occur simultaneously. After 5 days, a porous structure of platelike hydroxyapatite crystals appears on the surfaces of aggregated microspheres (Figure 5C), ultimately growing into a continuous coating (Figure 5D). The size of the aggregates depends on the initial concentration of microspheres in solution (Figure 3E), and mineral coatings were also observed on the surface of microsphere aggregates. Other analyses not presented here suggest that the efficiency of microsphere mineralization increases in conditions that promote microsphere aggregation. Our time-course analysis of mineral formation in solutions with higher concentrations of microspheres (Figure 5) also demonstrates that mineral nucleation can occur at the interface between aggregated microspheres (e.g., Figure 5B), which suggests that mineral formation can be facilitated by aggregation.

Interestingly, the mineral described here is analogous to the previous bioinspired mineral formation by He et al. (47), in which mineral was formed in the presence of dentin matrix protein. In addition, the morphology and composition of the mineral described here is similar to the apatite found in human bone and mineralized dentin (48), specifically a platelike nanostructure (Figures 1B and 8B), a hydroxyapatite phase (Figure 4A,C), and carbonate substitution (Figure 4B).

Dissolution is a key characteristic of hydroxyapatite coatings, particularly in potential drug delivery applications in which drug release has been shown to be dependent on the mineral dissolution rate. Therefore, in this study we

characterized dissolution of mineral coatings in TBS and in DMEM for 25 days.  $\text{Ca}^{2+}$  and  $\text{PO}_4^{3-}$  were gradually released from mineral coatings over time in TBS (Figure 6A), indicating that these coatings are less stable than pure, stoichiometric hydroxyapatite characterized in previous studies (13, 49, 50). The total amounts of  $\text{Ca}^{2+}$  and  $\text{PO}_4^{3-}$  released after 25 days in TBS were 15.87 and 10.26  $\mu\text{mol}$ , respectively, and the dissolution data indicated a Ca/P mol ratio in the range of 1.37–1.61 during the course of the 25 day study, indicating that the stoichiometry of the dissolving mineral may vary during the time course of dissolution. SEM images obtained at the end of the dissolution study also demonstrate a significant level of resorption of the mineral coating in TBS (Figure 6B) when compared to the initially coated microspheres (Figure 5D). The increased dissolution rate of these coatings when compared to pure hydroxyapatite coatings used in clinical applications can be explained by differences in crystallinity and carbonate substitution in the mineral. Fazan and Marquis (49) reported previously that the dissolution rate of plasma-sprayed hydroxyapatite coatings decreases with an increase in the degree of crystallinity of hydroxyapatite. In addition, it has been reported that the incorporation of sodium and carbonate ions into calcium phosphate minerals, such as hydroxyapatite, dramatically increases the dissolution rate (51), and our FTIR analyses clearly indicate the presence of carbonate ions in the mineral coatings (Figure 4B).

When mineral-coated microspheres were incubated in serum-free DMEM, there was a decrease in the cumulative amount of  $\text{PO}_4^{3-}$  in the dissolution media over time (Figure

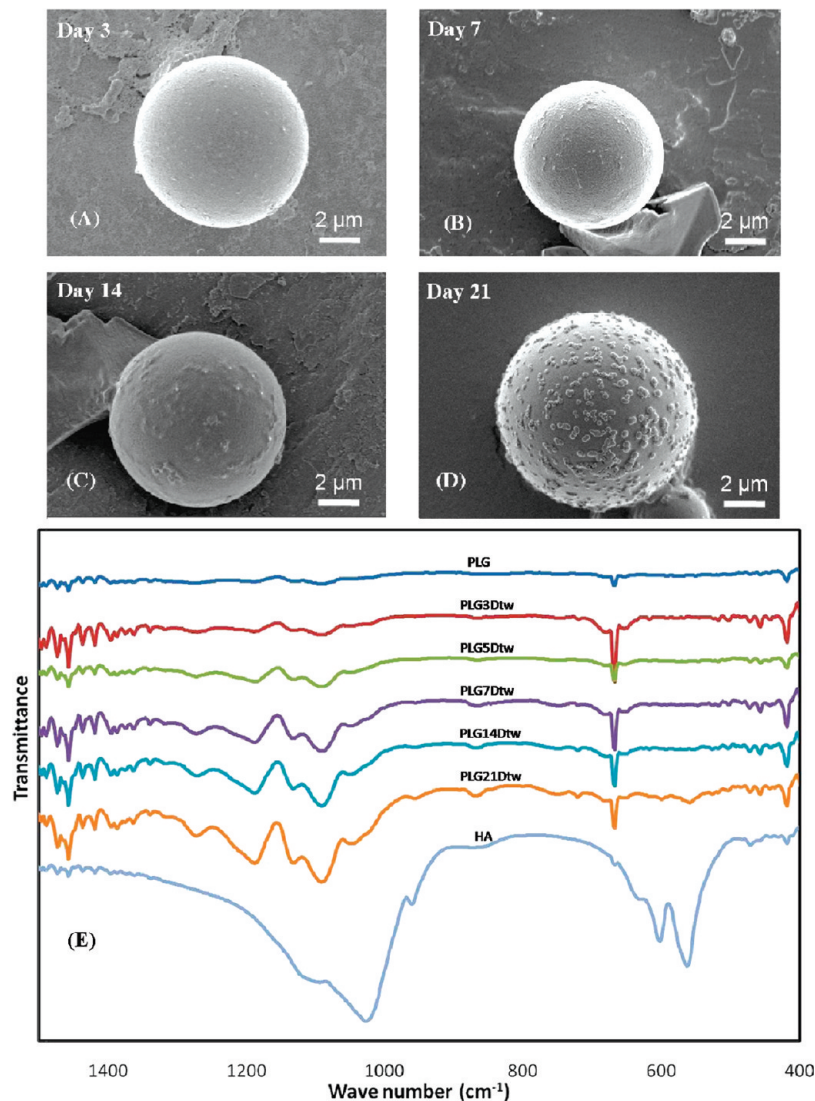


FIGURE 7. Effect of the surfactant (Tween 20) on the mineral formed on the PLG microsphere surfaces (A) after 3 days, (B) after 7 days, (C) after 14 days, and (D) after 21 days. (E) FTIR spectrum of PLG microspheres coated with mineral via a 21 day mSBF incubation in the presence of 0.1% (v/v) Tween20. A spectrum of commercial hydroxyapatite powder is included for comparison.

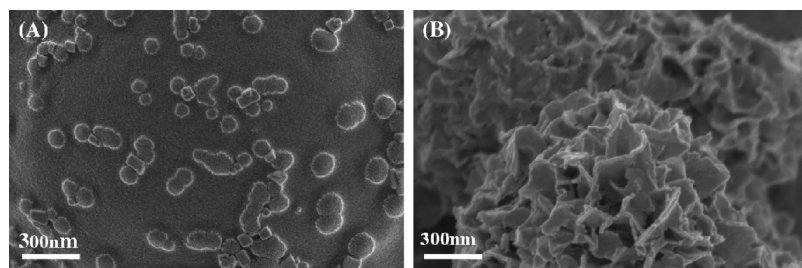


FIGURE 8. Nanometer-scale mineral morphology on the surface of microspheres formed (A) in the presence of 0.1% (v/v) Tween20 and (B) in the absence of Tween20.

6C), indicating that phosphate-containing mineral could possibly be reprecipitating on the surface of the mineral coating because of ion exchange between the carbonate-substituted hydroxyapatite and  $\text{PO}_4^{3-}$  in DMEM. This result suggests that the mineral coating could possibly serve as a substrate for nucleation of a calcium phosphate mineral component in the DMEM solution (Figure 6D). SEM images indicated that the morphology of the mineral coating after incubation in DMEM was similar to the morphology of the

mineral prior to incubation in DMEM, which suggests that if reprecipitation is occurring, it is resulting in the growth of a new mineral phase that is similar to the mineral phase grown initially. These results are consistent with previous work, in which hydroxyapatite was immersed in either TBS or a modified Hank's buffered saline (HBS) solution, which had an ionic composition similar to that of human plasma (50). In Tris buffer, there was no new mineral formed on the hydroxyapatite surface, while in the modified HBS solution,

an apatite coating was grown on the surface, and this new mineral coating had a morphology similar to that of the initial hydroxyapatite surface.

Previous studies have shown that drug release kinetics from the surface of hydroxyapatite are comparable to the hydroxyapatite mineral dissolution kinetics (13, 52). Therefore, the aggregation of the mineral-coated microspheres is likely to slow drug release because of the reduction in the surface area. Multiple recent studies have shown that protein release kinetics from the surface of hydroxyapatite strongly depend on the pH of the buffer medium (13, 53). For example, Matsumoto et al. demonstrated that enhanced mineral dissolution at low pH can lead to increased protein release. On the basis of these previous studies, we hypothesized that the composition and pH of buffer media would have an impact on the dissolution rate of the mineral. This hypothesis is supported by our data, indicating that the mineral dissolution rate is highly impacted when mineral-coated microspheres are incubated in either TBS or DMEM (Figure 6A,B). Ions were gradually released from the mineral over time in TBS (Figure 6C), whereas reprecipitation of the mineral occurred in phosphate-containing media (Figure 6D). This newly deposited mineral could interfere with drug diffusion out of the initially coated mineral, thereby slowing drug release.

**Effect of the Surfactant on Mineral Nucleation and Growth.** In order to modulate the aggregation of microspheres and observe its effect on mineral growth, Tween20 was added to the mSBF solution. Tween20 is a common surfactant used as a stabilizer in studies that measure drug release from PLG microspheres in vitro (54, 55). Mineral formation on the surface of microspheres is inhibited in the presence of Tween20 (Figure 7A–D) when compared to mineral formation in the absence of surfactant (Figures 1 and 3). FTIR spectra of the mineral formed in the presence of surfactant (Figure 7E) show peaks in the range of 650–800 and 1108–1414  $\text{cm}^{-1}$ , which indicate the presence of  $\text{PO}_4^{3-}$  similar to the mineral formed in the absence of surfactant (Figure 4B) and to synthetic hydroxyapatite (Figure 7E). The FTIR spectrum also included  $\text{CO}_3^{2-}$  peaks at 1410 and 1450  $\text{cm}^{-1}$ , which were absent in synthetic hydroxyapatite (Figure 7E). SEM images of the mineral formed in the presence of Tween20 show scattered islands of nanometer-scale mineral formation. Interestingly, the morphology of the mineral coatings in the presence of surfactant also differs significantly from the nanometer-scale platelike morphology apparent in the coatings formed without surfactant (Figure 8A,B). This result suggests that surfactant addition could be used in future studies as a mechanism to vary the mineral morphology on the microsphere surface, which could have implications for mineral degradation, binding of biological molecules, and biological activity. For example, a rough surface with a relatively well-distributed mineral coating has an increased surface area, and the corresponding increase in available binding sites could result in higher protein binding (13, 56).

#### IV. CONCLUSION

Mineral-coated PLG microspheres have been fabricated via a bioinspired process, which can be varied to modulate the resulting microsphere properties. XRD and FTIR spectra indicated that the coatings comprised a carbonate-substituted hydroxyapatite mineral with a porous, platelike nanoscale morphology. The size of the mineral-coated microspheres or mineral-coated microsphere aggregates can be controlled by varying the microsphere concentration in the mSBF solution. Hydroxyapatite coatings were degradable in TBS, and analysis of the calcium and phosphate release from the coatings indicated that the Ca/P molar ratio in the mineral is consistent with that of carbonate-substituted hydroxyapatite. The presence of a surfactant during the mineral growth process partially inhibited the formation of mineral and also significantly affected the morphology of the mineral. Taken together, these findings indicate that mineral-coated PLG microspheres can be synthesized in a controllable fashion using a bioinspired process. This material may be useful in a variety of applications that may benefit from the bulk properties of polymer microspheres and the surface properties of hydroxyapatite minerals, including drug delivery and biomolecule purification.

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**Supporting Information Available:** Graph indicating the percentage of aggregated microspheres. This material is available free of charge via the Internet at <http://pubs.acs.org>.

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